

- Diffusion mechanisms in crystals: vacancy, interstitial, interstitialcy
- Expressions for D:

Diffusivity of interstitial solute in bcc crystal

$$D = \frac{a^2}{6} \exp\left(-\frac{s^m}{k}\right) \exp\left(-\frac{h^m}{kT}\right) \quad f = 1$$

Self diffusion by vacancy mechanism

$$D = f a^2 \exp\left(-\frac{s_v^m + s_v^f}{k}\right) \exp\left(-\frac{h_v^m + h_v^f}{kT}\right) \quad f = 1 - \frac{2}{z}$$

Solute diffusion by vacancy mechanism: three-frequency model

- Background on defects in ionic crystals and Kröger-Vink notation

- **Atomistic models for diffusion coefficients - ionic crystals**

Intrinsic and extrinsic regimes in stoichiometric material

- **Diffusion in nonstoichiometric oxides; the oxygen sensor**

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- **Role of microstructure in diffusion: short-circuiting**

KPIM Chapter 9

• Cation diffusion in KCl with Ca impurities



Schottky equilibrium $K_S = [V_K] [V_{\text{Cl}}^\bullet] = \exp -\frac{g_S^f}{kT}$

Neutrality condition $[V_K] = [V_{\text{Cl}}^\bullet] + [\text{Ca}_K^\bullet]$

So we have $[V_K] = \frac{[\text{Ca}_K^\bullet]}{2} \left(1 + \sqrt{1 + \frac{4K_S}{[\text{Ca}_K^\bullet]^2}} \right)$

Intrinsic regime: impurity concentration is small or T is high and thus

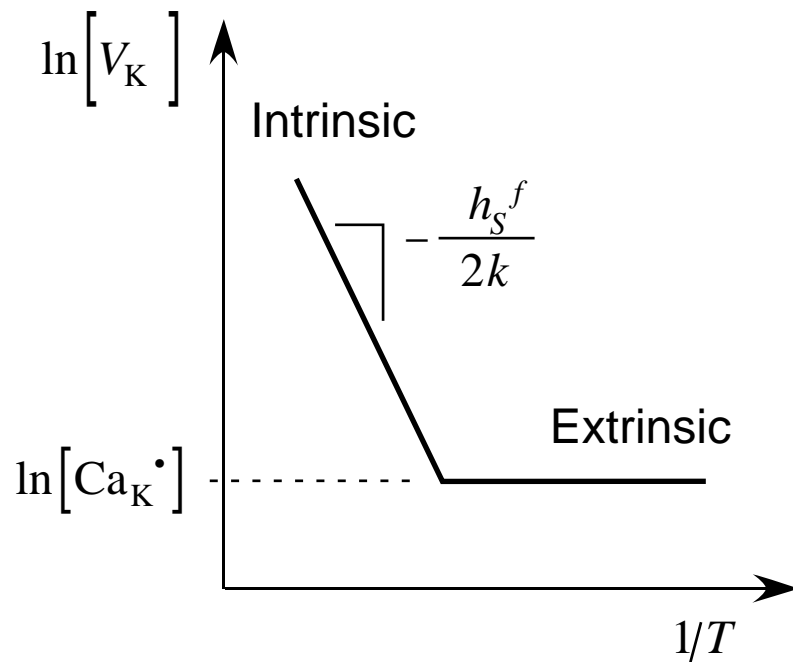
$$[V_K] = \sqrt{K_S}$$

Extrinsic regime: impurity concentration is high or T is low and thus

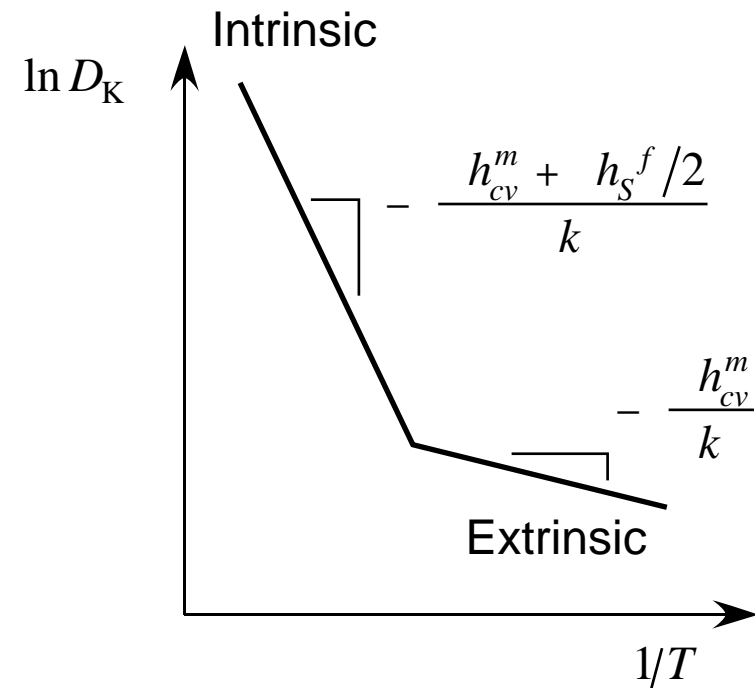
$$[V_K] = [\text{Ca}_K^\bullet]$$

Cation diffusion in KCl with Ca impurities, cont'd:

Temperature dependencies



Cation vacancy concentration

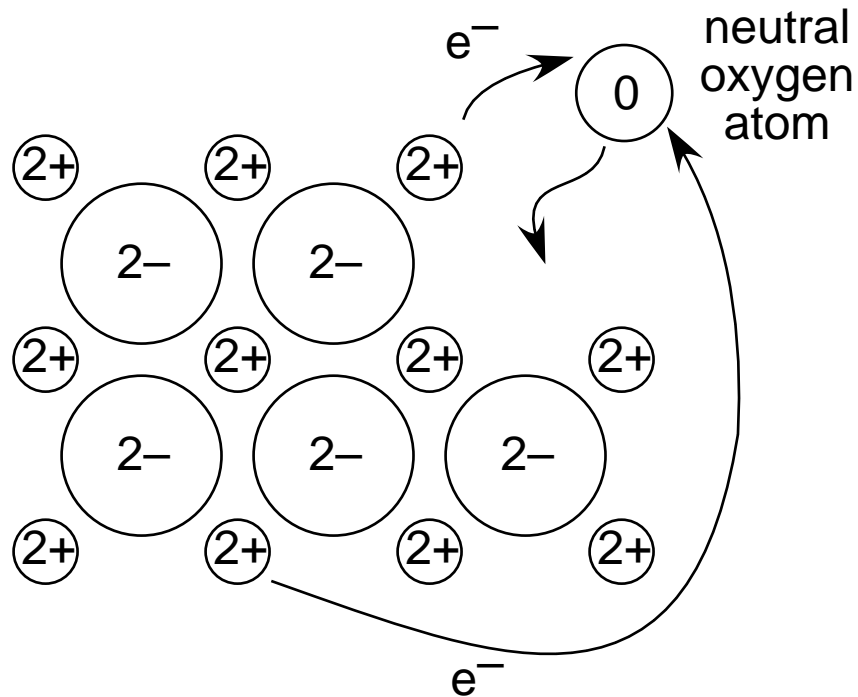


Cation diffusivity

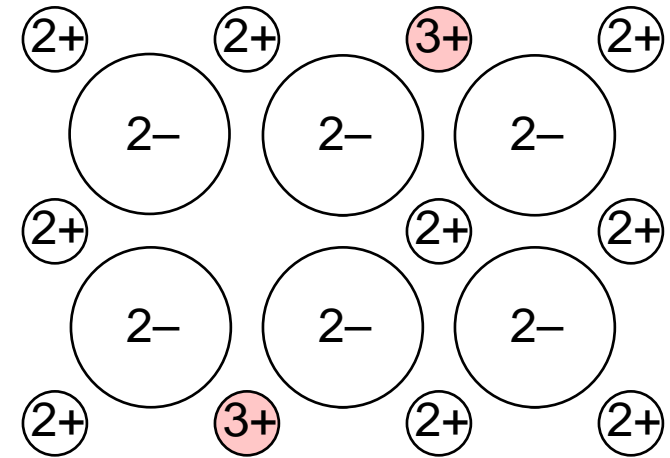
• Diffusion in nonstoichiometric oxides

— e.g., cation diffusion in FeO (note various valence states of Fe are possible)

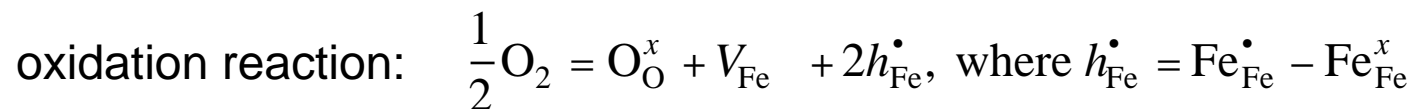
FeO can be *oxidized* to make a *cation deficient* oxide Fe_{1-x}O



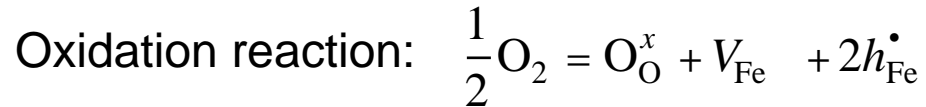
Oxygen atom approaches, attracts two electrons from Fe²⁺ and gets ionized.



Oxygen-deficient crystal. Note cation vacancy and two ferric cations.



• **Diffusion in nonstoichiometric FeO, cont'd:**



Equilibrium constant for this reaction:
$$K_{eq} = \frac{[V_{Fe}][h_{Fe}^\bullet]^2}{\sqrt{p_{O_2}}} = \exp -\frac{G}{kT}$$

Charge neutrality: $2[V_{Fe}] = [h_{Fe}^\bullet]$

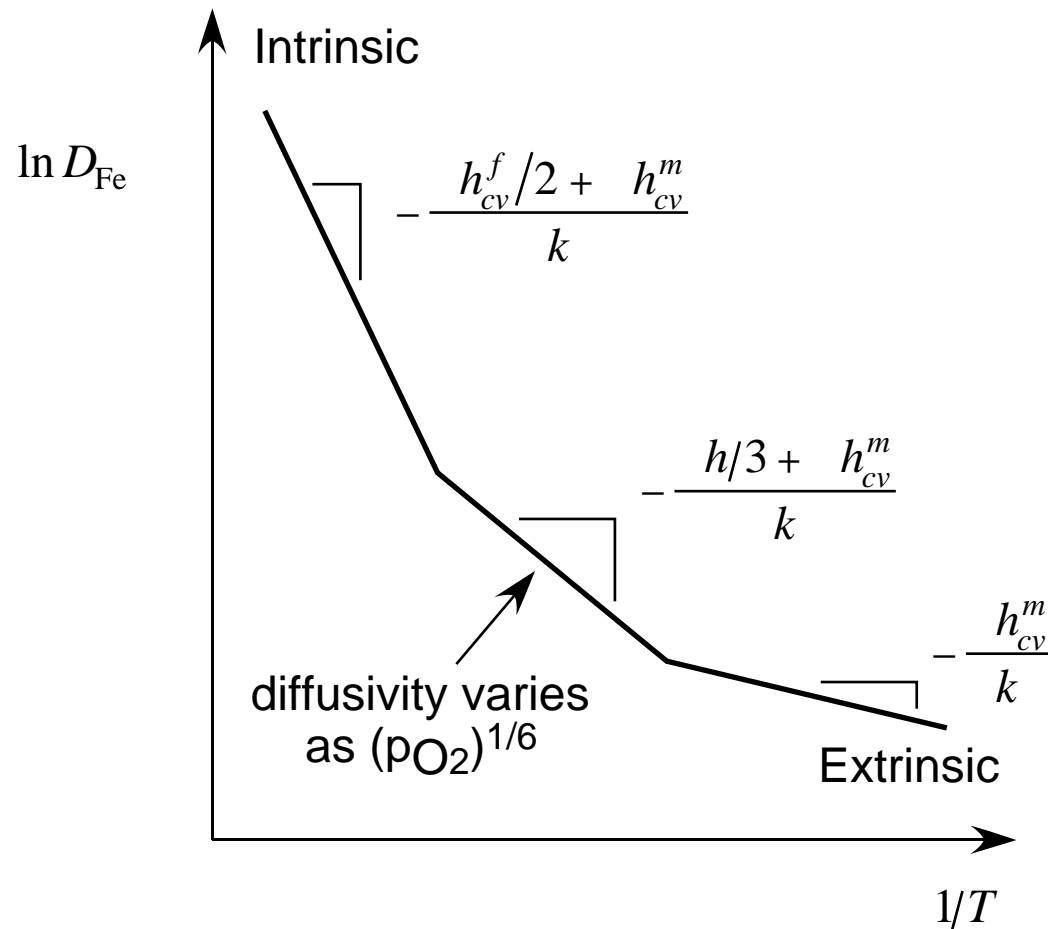
Cation vacancy concentration: $[V_{Fe}] = (1/4)^{1/3} \exp -\frac{G}{3kT} (p_{O_2})^{1/6}$

Diffusion in impure FeO will have *three* regimes:

1. High T , low oxygen pressure, dominated by Schottky defect equilibria
2. High oxygen pressure, dominated by oxidation reaction
3. Low T , low oxygen pressure, dominated by extrinsic impurities

• Diffusion in nonstoichiometric FeO, cont'd:

Cation diffusivity Arrhenius plot



Position of middle segment will depend on oxygen pressure, hence this region will not be observable at low oxygen pressures or high impurity contents

• Diffusion in ceramics: relevance

Microstructural processes

Densification of powder compacts by sintering

Creep deformation at high temperatures

Grain growth

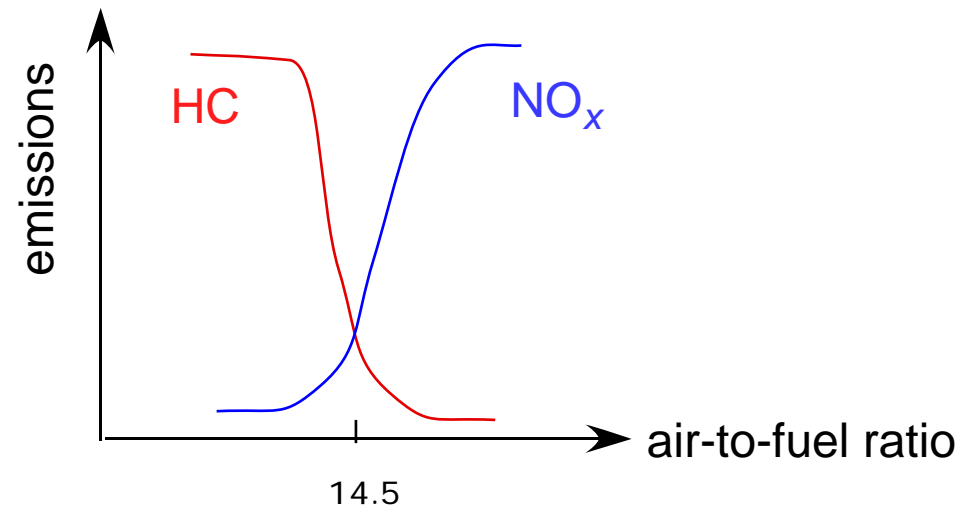
Solid–solid transformation kinetics, including oxidation of metals

Electrical conduction

Ionic conductors for chemical and gas sensors, solid electrolytes, fuel cells

The automotive oxygen sensor

sensors



The automotive oxygen sensor, cont'd:

TiO₂-based, works on *electronic* conduction that has pO₂ dependence

ZrO₂-Y₂O₃-based, works on electrochemical potential difference between
exhaust stream and ambient air

References for additional study:

KPIM Chapter 8

Chiang et al., *Physical Ceramics*, 1997.

Section 2.1 on Point Defects

Section 3.2 on Atomistic Diffusion Processes

Section 3.3. on Electrical Conductivity

Oxygen sensors, pp. 142–146 and pp. 234-235

Allen and Thomas, *The Structure of Materials*, 1999

Section 5.1.6 on Point Imperfections in Ionic Crystals